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# Effects of dopants on grain growth of nano-sized BaTiO<sub>3</sub> powders prepared by citric acid-assisted spray pyrolysis

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Nano-sized BaTiO<sub>3</sub> powders doped with Y, Eu and La components were prepared by ultrasonic spray pyrolysis from a spray solution with citric acid. The precursor BaTiO<sub>3</sub> powders had a large size, fractured and hollow structures irrespective of the type of dopant. The calcined BaTiO<sub>3</sub> powders with and without dopants had nanometre sizes after a simple milling by hand using an agate mortar. The mean size of La-doped BaTiO<sub>3</sub> powders measured from the transmission electron microscope (TEM) images was 38 nm at a calcination temperature of 700 °C. The mean crystallite sizes of the doped and undoped BaTiO<sub>3</sub> powders changed from 20 to 33 nm according to the type of dopant at a calcination temperature of 700 °C. An effective grain growth inhibition took place with the doped BaTiO<sub>3</sub> ceramics. The average grain sizes of La and Eu-doped BaTiO<sub>3</sub> ceramics were each 380 and 400 nm at a sintering temperature of 1280 °C.

Key words: Spray pyrolysis, Nano particle, Barium titanate.

## Introduction

With further developments in electronics, the miniaturization of electronic parts has proceeded quickly and the desire to increase the capacity of multilayer ceramic capacitors (MLCC) has also become more and more urgent [1, 2]. For MLCC fabrication, a dielectric layer thickness approaching 1  $\mu$ m that can be co-fired with electrode materials is desired [3, 4]. For this reason, current MLCC industries prefer to use BaTiO<sub>3</sub> powders that are nano-sized, structured with regular morphologies, have a narrow size distribution, and show no excessive grain growth after sintering.

Some additives as dopant have been reported to effectively prevent discontinuous grain growth [5, 6]. Effective grain-growth inhibition for donor-doping occurs between 0.20 and 0.40 mol% with materials such as  $La_2O_3$  [7, 8] and  $Sb_2O_3$  [9]. The critical concentration has been conveniently termed the grain-growth inhibition threshold (GGIT) [6]; beyond which, grain growth in donor-doped BaTiO<sub>3</sub> ceramics is effectively hindered, the microstructure refined and semiconductivity sharply reduced. Donor-doped BaTiO<sub>3</sub> powders with a nanometre size have mainly been prepared by liquid phase reaction methods. Cui et al. prepared Nb-doped BaTiO<sub>3</sub> nanocrystalline powders and ceramics by a simple solgel process [10]. Nb-doped BaTiO<sub>3</sub> ceramics calcined at 1000 °C for 2 h and sintered at 1250 °C for 2 h had a regular microstructure with the maximum dielectric constant of 10,298. Zhao et al. prepared nano-sized  $BaTiO_3$  doped with La by sol-gel process [11]. The powder size and grain size decreased with adding dopants.

Nano-sized BaTiO<sub>3</sub> powders have also been prepared by gas phase reaction methods. Lee et al. prepared nanosized BaTiO<sub>3</sub> powders by spray pyrolysis from a spray solution containing an organic additive [12]. Xia et al. developed a new preparation strategy for spray pyrolysis to efficiently produce nano-sized BaTiO<sub>3</sub> powders, the so called 'salt-assisted' spray pyrolysis process [13]. Submicrometre size powders prepared by salt-assisted spray pyrolysis changed into nano-sized BaTiO<sub>3</sub> powders with non-agglomeration characteristics after a repeated washing process using distilled water. However, the donor-doped BaTiO<sub>3</sub> powders were not well studied to the gas phase reaction methods.

In this study, donor-doped  $BaTiO_3$  powders with a nanometre size were prepared by ultrasonic spray pyrolysis from a spray solution with citric acid. The effects of the type of donor on the characteristics of nano-sized  $BaTiO_3$  powders, such as mean size, crystallite size, and morphology, were investigated. The sintering characteristics of the donor-doped  $BaTiO_3$  powders were also investigated.

#### **Experimental Procedure**

The spray pyrolysis system consisted of a droplet generator, a quartz reactor, and a teflon bag filter. A 1.7 MHz ultrasonic spray generator having six vibrators was used to generate a large amount of droplets, which were carried into the high temperature tubular reactor by a carrier gas. The flow rate of air used as a carrier gas was 45 Lminute<sup>-1</sup>. Droplets and powders evaporated, decomposed, and/or crystallized in the quartz reactor. The

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length and diameter of the quartz reactor were 1200 and 50 mm, respectively.

Barium carbonate and titanium tetra-iso-propoxide (TTIP) were used as starting materials to prepare BaTiO<sub>3</sub> powders. The starting materials were added into a mixed solution of water and nitric acid to form a clear solution. The concentration of Ba and Ti was fixed at 0.1 M. The concentration of citric acid used as an organic additive was 0.4 M. Various doping materials such as Y<sub>2</sub>O<sub>3</sub>, Eu<sub>2</sub>O<sub>3</sub>, and  $La_2O_3$  were added into the spray solution. The doping concentration of Y, Eu, and La components was 0.02 mol% of BaTiO<sub>3</sub>. The as-prepared powders obtained by spray pyrolysis were calcined at 700 °C for 2 h in an air atmosphere. The calcined powders were pelletized at  $2.45 \times 10^7$  Pa pressure into 15 mm diameter pellets. The pellets were then sintered at 1280 °C for 5 h and cooled naturally to room temperature by turning off the furnace power.

The crystal structures of the doped-BaTiO<sub>3</sub> calcined at 700 °C for 2 h were investigated using X-ray diffraction (XRD, RIGAKU, D/MAX-RB) with Cu K $\alpha$  radiation ( $\lambda = 1.5418$  Å). The mean crystallite sizes of the BaTiO<sub>3</sub> powders were calculated using Scherrer's equation. The morphological characteristics of the powders and pellets were investigated using scanning electron microscopy (SEM) and transmission electron microscopy (TEM). The calcined powders were milled by hand using an





**Fig. 1.** SEM images of BaTiO<sub>3</sub> precursor powders; (a) undoped, (b) La-doped.

agate mortar for the preparation of TEM samples.

## **Results and Discussion**

In spray pyrolysis, hollow powders can be formed when a solute concentration gradient is created during evaporation of droplet. The solute precipitates first at the more highly supersaturated surface if sufficient time is not available for solute diffusion in the droplet. Thus, the morphology of powders is strongly influenced by the preparation conditions in spray pyrolysis. The organic additives added into the spray solution also affect the morphology of the powders prepared by spray pyrolysis by controlling the drying and decomposition stages of the droplets or powders. In this study, BaTiO<sub>3</sub> powders with a hollow and porous morphology were prepared by spray pyrolysis from the spray solution with citric acid under severe preparation conditions. The residence time of the powders inside the hot wall reactor was 0.6 s. Figure 1 shows SEM images of precursor BaTiO<sub>3</sub> powders with and without La dopant.



**Fig. 2.** SEM images of calcined BaTiO<sub>3</sub> powders; (a) undoped (b) Y-doped (c) Eu-doped and (d) La-doped.

The precursor  $BaTiO_3$  powders with and without the dopant had a large size, with fractured and hollow structures irrespective of the type of dopant. Gas evolution generated by the decomposition of the citric acid caused the powders to be hollow and with thin wall structures. The mechanism of formation of powders with hollow and thin wall structures in spray pyrolysis from a spray solution with organic additives has been described in a previous paper [12].

Figure 2 shows SEM images of the doped and undoped BaTiO<sub>3</sub> powders calcined at a temperature of 700 °C for



Fig. 3. TEM images of calcined  $BaTiO_3$  powders; (a) Y-doped, (b) Eu-doped, (c) La-doped.

2 h. The morphologies of the doped and undoped BaTiO<sub>3</sub> powders before (left) and after (right) the milling process are shown. The calcined powders were milled by hand using an agate mortar. The spherical shape and several micrometre size of the precursor powders were maintained after calcination. The micrometre-sized BaTiO<sub>3</sub> powders had an aggregated morphology of primary powders with a nanometre size irrespective of the dopant. The neck strength between the nano-sized primary powders was weak. Therefore, it was easy to disintegrate the nano-structured BaTiO<sub>3</sub> powders into non-aggregated powders with a nanometre size by the simple milling process. The spherical shape and micrometre sizes of the calcined BaTiO<sub>3</sub> powders disappeared after this simple milling process. The microstructures of the doped BaTiO<sub>3</sub> powders calcined at a temperature of 700 °C are shown in Fig. 3. The calcined BaTiO<sub>3</sub> powders had slightly the aggregated morphology of the primary powders with a nanometre size. The mean size of La-doped BaTiO3 powders measured from the TEM images was 38 nm.

Figure 4 shows the XRD patterns of the doped and undoped BaTiO<sub>3</sub> powders calcined at a temperature of 700 °C for 2 h. The calcined BaTiO<sub>3</sub> powders had a pure cubic phase regardless of the type of dopant. However, the crystallinities of the powders were affected by the type of dopant. The mean crystallite size of the doped and undoped BaTiO<sub>3</sub> powders changed from 20 to 33 nm according to the type of dopant. The mean crystallite size of the powders were calculated using Scherrer's equation from the width of the XRD peaks located at  $2\theta = 31.7^{\circ}$ .

The type of dopant affected the characteristics of the sintered BaTiO<sub>3</sub> pellets. Figure. 5 shows SEM images of the surfaces of sintered BaTiO<sub>3</sub> ceramics. From the SEM images, it can be seen that the grains have grown well with a regular and dense crystal shape. Interestingly, an effective grain growth inhibition took place with the doped BaTiO<sub>3</sub> ceramics, while the pure BaTiO<sub>3</sub> has a large grain size of 3-4  $\mu$ m. This result is similar to the



Fig. 4. X-ray diffraction patterns of calcined BaTiO<sub>3</sub> powders.



Fig. 5. Microstructures of BaTiO<sub>3</sub> pellets; (a) undoped, (b) Y-doped, (c) Eu-doped, (d) La-doped.

Ti-excess BaTiO<sub>3</sub> powders with donor oxides [7, 14-16]. Suppressed growth microstructures below 500 nm are shown in Fig. 5(c), (d). The average grain sizes of La and Eu-doped BaTiO<sub>3</sub> ceramics, estimated from SEM images, were 380 and 400 nm respectively.

#### Conclusions

The type of dopant affected the characteristics of nanosized BaTiO<sub>3</sub> powders prepared by ultrasonic spray pyrolysis, such as the mean size, mean crystallite size, and morphology. The type of dopant also affected the characteristics of the sintered BaTiO<sub>3</sub> ceramics. The grains of BaTiO<sub>3</sub> ceramics grew well densely with a regular shape. The pure BaTiO<sub>3</sub> ceramic had a large grain size of 3-4  $\mu$ m. However, doped BaTiO<sub>3</sub> ceramics had fine grain sizes below 500 nm and regular microstructures after sintering at a temperature of 1280 °C in an air atmosphere.

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