

Photocatalytic activity of ZrO₂-doped TiO₂ catalysts prepared by a surfactant-assisted templating method

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ZrO₂-doped TiO₂ nanocrystals were prepared by a surfactant-assisted templating method (SATM) under a sol-gel process. Mesoporous anatase-type ZrO₂-doped TiO₂ nanocrystals with a surface area of 40-80 m²/g were obtained. A small amount of ZrO₂ addition to TiO₂ nanocrystals increased the thermal stability of the anatase phase of TiO₂. By using a photocatalytic reaction of triiodide (I₃⁻) ions formation in KI solution by UV-irradiation, it was found that the 0.5 mol% ZrO₂-doped TiO₂ nanocrystals possessed the best photocatalytic activity of the TiO₂-ZrO₂ nanocrystals tested in this study. The effects of calcination temperature and zirconia content on the photocatalytic activity of titania-based nanocrystals were investigated.

Key words: Titania, Photocatalytic activity, Surfactant-assisted templating method, Anatase

Introduction

Nanostructured titania (TiO₂) materials have been the subject of a great deal of research because of their properties and applications in photocatalysis, solar energy conversion, electronic devices, cosmetics, environmental purification, etc. [1-4]. TiO₂ has been widely utilized as a photocatalyst because it is relatively safe, inexpensive and stable to photocorrosion. Oxidation cleavages, condensation, polymerization, geometric and valence isomerization, and substitutions, have been accomplished through photocatalysis with TiO₂ [5-8]. Recently, Karunakaran *et al.* [9] have been studied the photocatalytic activity of TiO₂ by photooxidation of iodide ion. The results showed the catalytic efficiency for the formation of iodine in a TiO₂ photocatalyst aqueous ethanol suspension. The efficiency of the catalyst depends mainly on the intensity of absorbed photons during illumination, the formation of electron-hole pairs and recombination rates, the charge transfer rate, and the reaction conditions [10, 11]. The photocatalytic activity has been improved by optimizing the nanostructure of TiO₂, using various processing routes, such as inert gas condensation [12], a sol-gel method [13, 14], and hydrothermal processing [15]. The photocatalytic activity can also be altered by doping with a transition metal oxide such as ZrO₂ [16], WO₃ [17], etc.; the addition of metal or metal oxide enhances the thermal stability of the anatase phase, and increases the surface area and surface acidity, resulting in improved photocatalytic activity [18-20].

In our previous study, mesoporous TiO₂ and ZrO₂-doped TiO₂ nanocrystals have been successfully prepared by a surfactant-assisted templating method (SATM) [21, 22]. By using this method, mesoporous materials with controlled and well-defined pore-structures can be fabricated. The addition of ZrO₂, improved the thermal stability of the metastable anatase phase, which has a better photocatalytic activity than the rutile phase [23]. In this paper, more detailed and further experimental results on the photocatalytic activity of mesoporous ZrO₂-doped TiO₂ nanocrystals will be presented.

Experimental Procedure

All chemicals were analytical grade and used without further purification. The ZrO₂-doped TiO₂ nanocrystals were prepared by SATM [21, 22, 24]. Tetra(*i*-propyl) orthotitanate (TIPT, Tokyo Chemical Industry Co., Japan), acetylacetone (ACA, Nacalai Tesque, Inc., Japan), and laurylamine hydrochloride (LAHC, Tokyo Chemical Industry Co., Japan) were used as TiO₂ precursor, modifying agent, and templating structures, respectively. Zirconyl nitrate hydrate (ZrO(NO₃)₂, Aldrich) was used as ZrO₂ precursor. An LAHC solution (0.1 M) containing the appropriate amount of zirconia precursor was mixed with the equimolar mixture of TIPT and ACA and magnetically stirred at room temperature for 1 h. The resulting mixture was further stirred in an oven at 40°C for 24 h. Then, the homogeneous sol was kept sealed at 80°C. After 72 h, the gel obtained was washed with 2-propanol (Nacalai Tesque, Inc., Japan) and dried at 80°C overnight. Finally, the templates were removed from the TiO₂ framework by a calcination process.

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Characterization of titania nanocrystals

Transmission electron microscopy (TEM) and selected area electron diffraction (SAED) were performed on a transmission electron microscope (JEOL JEM-200CX) at 200 kV. The X-ray diffraction (XRD) analysis was performed on a micro X-ray diffractometer (Model RINT-2100, Rigaku) with Cu-K α radiation ($\lambda=0.154$ nm) at 40 kV and 40 mA, and a scan rate of 2°(2 θ)/minute. The N₂ adsorption-desorption isotherms were obtained with a nitrogen adsorption apparatus (BELSORP18 PLUS). The powders were further degassed under vacuum at 200°C for 2 h before measurements to evacuate the physisorbed moisture. Diffuse reflectance spectra were recorded on a UV-vis spectrophotometer (UV-2450 Spectrophotometer, Shimadzu). BaSO₄ was used as the reference powder. The collected relative reflection intensity ($R_{\infty}=R_{\text{sample}}/R_{\text{reference}}$) was transformed into $F(R_{\infty})$ by using the Kubelka-Munk function $F(R_{\infty}) = (1-R_{\infty})^2/(2R_{\infty})$ [25]. All spectra were plotted in terms of $F(R_{\infty})$ vs. wavelengths. The extrapolation of the absorption edge to the wavelength axis gives the photon energy (E_{phot}) as $E_{\text{phot}} = (1239/\lambda)$ eV, where λ is wavelength (in nm). The value of E_{phot} corresponds to the band gap energy [26]. The photocatalytic activity was investigated by adding 50 mg of sample powder into 10 ml, 0.2 M KI aqueous solution in a cylinder reactor. The suspension was magnetically stirred and photoirradiated with a UV lamp (15 W, Vilber Lourmat VL-115L, with a maximum emission at about 365 nm). The concentration of liberated I₃⁻ ions in the clear supernatant after diluting ten times was monitored at regular time intervals by determining the absorbance at 288 nm, using an UV-vis spectrophotometer (Shimadzu UV 2450). The molar extinction coefficient (ϵ_{max}) was determined to be 4.0×10^4 (cm·mol/l)⁻¹. The experiment was repeated for chosen samples as well as for the commercially-available titania nanomaterials, ST-01 (Ishihara Sangyo Kaisha, Ltd., Japan), which were tested for the sake of comparison. The commercial TiO₂ nanomaterial was used in the as-received state without further treatment before investigations. No I₃⁻ formation was observed when the experiments were conducted in the dark or in the absence of the TiO₂ specimens.

Results and Discussion

A TEM image and a SAED pattern of the 0.5 mol% ZrO₂-doped TiO₂ nanocrystals are given in Fig. 1. As can be seen in this figure, aggregates of nanoparticles were observed. The size of the anatase-type TiO₂ nanocrystals decreased from about 10-20 nm for undoped TiO₂ nanocrystals prepared by SATM [21], to about 7-15 nm for the 0.5 mol% ZrO₂-doped TiO₂ nanocrystals. It is obvious that the crystallite growth is suppressed by the presence of ZrO₂ as a guest metal oxide. Figure 2 gives XRD patterns of the 0.5 mol% ZrO₂-doped TiO₂

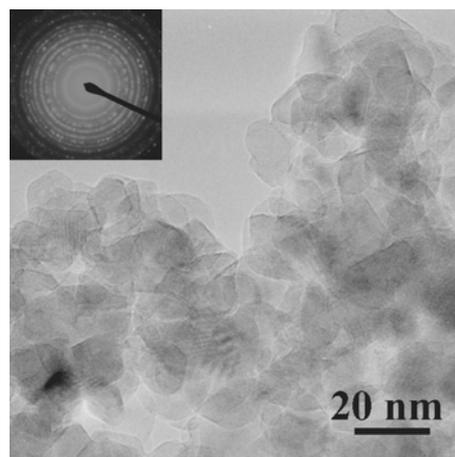


Fig. 1. TEM image of the 0.5 mol% ZrO₂-doped TiO₂ nanocrystals (inset; SAED of nanocrystals) calcined at 500°C.

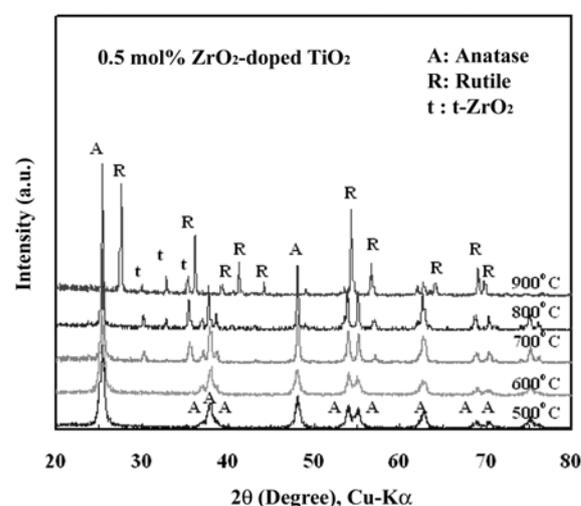


Fig. 2. XRD patterns of the 0.5 mol% ZrO₂-doped TiO₂ nanocrystals prepared by SATM.

nanocrystals derived from [ACA]/[Ti+Zr]=1, [LAHC]/[Ti+Zr]=0.25, and calcined at different temperatures (500°C-900°C) in air. It has been reported that fine anatase-type TiO₂ transforms to the rutile phase after calcination at 550°C-650°C [27, 28]. The XRD patterns revealed that the addition of ZrO₂ to TiO₂ tends to inhibit the phase transformation from the anatase to the rutile phase of TiO₂. Figure 3 shows representative examples of UV-vis diffuse reflectance spectra of the undoped TiO₂ and the 0.5 mol% ZrO₂-doped TiO₂ nanocrystals compared with that of ST-01 (equi-axed fine anatase particles, consisted of anatase). The absorption edge of the undoped TiO₂ (mesoporous anatase-type) by SATM calcined at 500°C occurs at ~380 nm, and its band gap energy is estimated to be about 3.26 eV. Meanwhile, the absorption edge of the 0.5 mol% ZrO₂-doped TiO₂ is slightly shifted by ~8 nm to a longer wavelength in comparison with that of the undoped TiO₂ nanocrystals. By adding ZrO₂, the value of the band gap is found to become slightly

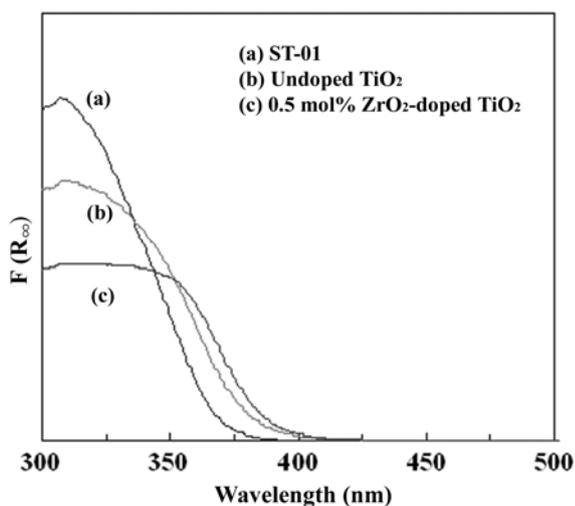


Fig. 3. UV-vis diffuse reflectance spectra of (a) ST-01, (b) undoped TiO_2 by SATM, and (c) 0.5 mol% ZrO_2 -doped TiO_2 nanocrystals by SATM.

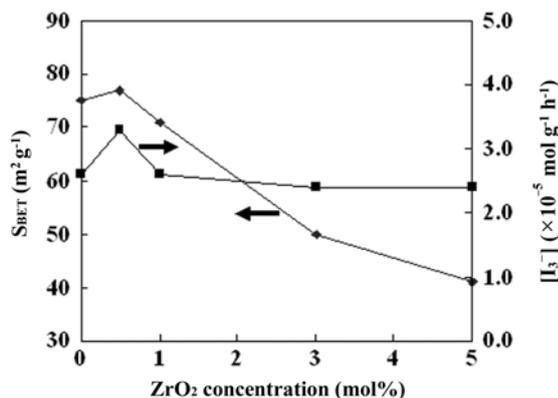


Fig. 4. Surface area and photocatalytic activity of the ZrO_2 -doped TiO_2 nanocrystals (calcination temperature: 500°C) for different dopant concentration (in mol%).

lower than that of the undoped TiO_2 , and with a slight decrease in energy band gap value.

The photocatalytic activity of the above mentioned mesoporous ZrO_2 -doped TiO_2 nanocrystals were evaluated based on the formation of triiodide (I_3^-) species in a suspension of the TiO_2 -based photocatalyst in KI aqueous solution after UV-irradiation. The surface area and the photocatalytic activity (estimated as a concentration of I_3^- in 10 ml of 0.2 M KI solution per unit mass of TiO_2 powder), for different dopant concentrations are depicted in Fig. 4. These results revealed that the photocatalytic activity and S_{BET} increased up to 0.5 mol% ZrO_2 doping and decreased for further dopant concentrations. It is considered that the 0.5 mol% ZrO_2 -doped TiO_2 should have improved photocatalytic activity due to the rather high S_{BET} and small crystallite size in the presence of dopant species [7]. The photocatalytic reaction occurs on the surface of the TiO_2 catalyst during UV illumination and the recombination of photogenerated electrons and holes is very

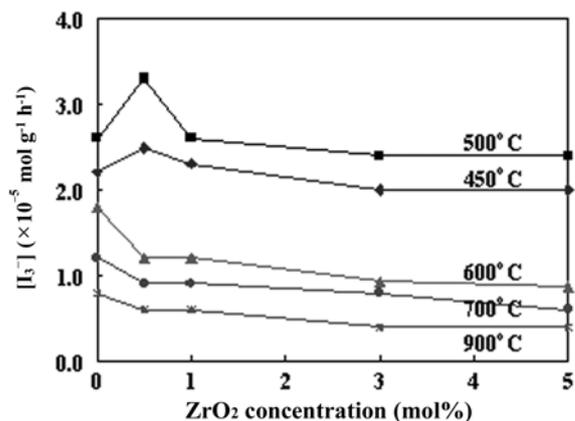


Fig. 5. The effect of calcination temperatures (450°C - 900°C) on the photocatalytic activity of 0.5 mol% ZrO_2 -doped TiO_2 nanocrystals.

fast [29]. The small amount of ZrO_2 may inhibit the electron-hole pair recombination and cause the enhancement of the photocatalytic activity. Although the photocatalytic activity of the 0.5 mol% ZrO_2 -doped TiO_2 nanocrystals was $\sim 30\%$ lower than that of ST-01 tested under similar condition ($5.0 \times 10^{-5} \text{ mol} \cdot \text{g}^{-1} \cdot \text{h}^{-1}$), the small amount of ZrO_2 addition itself should be a promising way to improve the photocatalytic activity of TiO_2 . The ZrO_2 content dependence of the photocatalytic activity for the ZrO_2 -doped TiO_2 with different calcination temperatures (500°C - 900°C) is shown in Fig. 5. It is obvious that the 500°C calcined nanocrystals exhibited the highest photocatalytic activity of all the ZrO_2 contents, which was attributed to the higher S_{BET} , and crystallinity. At calcination temperatures of 450°C - 500°C , the 0.5 mol% doping showed maximum values. A very small amount of ZrO_2 doping may contribute to enhance the crystallinity of TiO_2 [16] and inhibit the electron-hole recombination, and hence improve the photocatalytic activity. At temperatures higher than 500°C , all ZrO_2 -doped TiO_2 nanocrystals exhibited lower photocatalytic activity than that of undoped TiO_2 , due to the existence of the tetragonal ZrO_2 phase found on bulk powder [30], as shown in Fig. 2.

Conclusions

For mesoporous TiO_2 nanocrystals prepared by SATM, a small amount of ZrO_2 doping (0.5 mol%) improved the photocatalytic activity. ZrO_2 doping played a role in retarding the phase transformation from the anatase phase to the rutile phase of TiO_2 nanocrystals during calcining at various calcination temperatures and increased crystallinity with a reduced crystallite size. The photocatalytic activity of synthesized ZrO_2 -doped TiO_2 nanocrystals greatly depends on the concentration of ZrO_2 and calcination conditions. The small amount ZrO_2 -doped TiO_2 nanocrystals will be a key technique to have a high

surface area and small crystallite to enhance the photocatalytic activity of TiO₂.

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References

1. W. Guo, Z. Lin, X. Wang, and G. Song, *Microelec. Eng.* 66 (2003) 95-101.
2. M.R. Hoffmann, S.T. Martin, W. Choi, and D.W. Bahnemann, *Chem. Rev.* 95 (1995) 69-96.
3. A. Fujishima and K. Honda, *Nature* 238 (1972) 37-38.
4. P.A. Mandelbaum, A.E. Regazzoni, M.A. Blesa, and S.A. Bilmes, *J. Phys. Chem. B* 103 (1999) 5505-5511.
5. M.A. Fox and M.T. Dulay, *Chem. Rev.* 93 (1993) 341-357.
6. L. Saadoun, J.A. Ayllón, J. Jiménez-Becerril, J. Peral, X. Domènech, and R. Rodríguez-Clemente, *Appl. Catal. B: Environ.* 21 (1999) 269-277.
7. Y. Zhang, G. Xiong, N. Yao, W. Yang, and X. Fu, *Catal. Today* 68 (2001) 89-95.
8. A. Di Paola, G. Marci, L. Palmisano, M. Schiavello, K. Uosaki, S. Ikeda, and B. Ohtani, *J. Phys. Chem. B* 106 (2002) 637-645.
9. C. Karunakaran, S. Senthilvelan, S. Karuthapandian, and K. Balaraman, *Catal. Commun.* 5 (2004) 283-290.
10. J.C. Yu, J. Lin, D. Lo, and S.K. Lam, *Langmuir* 16 (2003) 7304-7308.
11. G. Sivalingam, K. Nagaveni, M.S. Hegde, and G. Madras, *Appl. Catal. B: Environ.* 45 (2003) 23-38.
12. H. Hahn and R.S. Averback, *Nanostruc. Mater.* 1 (1992) 95-100.
13. H. Yin, Y. Wada, T. Kitamura, S. Kambe, S. Murasawa, H. Mori, T. Sakata, and S. Yanagida, *J. Mater. Chem.* 11 (2001) 1694-1703.
14. M.K. Akhtar, S. Vermury, and S.E. Pratsinis, *Nanostruc. Mater.* 4 (1994) 537-544.
15. Yu. V. Kolen'ko, V. D. Maximov, A. A. Burukhin, V. A. Muhanov, and B. R. Churagulov, *Mater. Sci. Eng.: C* 23(6-8) (2003) 1033-1038.
16. K.Y. Jung and S.B. Park, *Mater. Lett.* 58 (2004) 2897-2900.
17. Y.R. Do, W. Lee, K. Dwight, and A. Wold, *J. Solid State Chem.* 108 (1994) 198-201.
18. X. Fu, L.A. Clark, Q. Ying, and M.A. Anderson, *Environ. Sci. Technol.* 30 (1996) 647-653.
19. R.N. Viswanath and S. Ramasamy, *Colloids Surf. A* 133 (1998) 49-56.
20. D. Das, H.K. Mishra, A.K. Dalai, and K.M. Parida, *Appl. Catal. A: Gen.* 243 (2003) 271-284.
21. S. Sakulkaemaruehai, Y. Suzuki, and S. Yoshikawa, *J. Ceram. Soc. Jpn.*, 112 (2004) 547-552.
22. S. Sakulkaemaruehai, Y. Suzuki, and S. Yoshikawa, *J. Jpn. Soc. Powder Powder Metallurgy*, 51 (2004) 789-794.
23. J. Augustynski, *Electrochim. Acta* 38 (1993) 43-46.
24. M. Adachi, Y. Murata, H. Harada, and S. Yoshikawa, *Chem. Lett.* (2000) 942-943.
25. G. Kortum, in “*Reflectance Spectroscopy*” (Springer-Verlag, 1969) p.22.
26. K.M. Reddy, S.V. Manorama and A.R. Reddy, *Mater. Chem. Phys.* 78 (2002) 239-245.
27. S.R. Yoganarasimhan and C.N.R. Rao, *Trans. Faraday Soc.* 58 (1962) 1579-1589.
28. J.C.S. Wu and C.-Y. Yeh, *J. Mater. Res.* 16 (2001) 615-620.
29. A.L. Linsebigler, G. Lu, and J.T. Yates, Jr., *Chem. Rev.* 95 (1995) 735-758.
30. M. Hirano, C. Nakahara, K. Ota, O. Tanaike, and M. Inagaki, *J. Solid State Chem.* 170 (2003) 39-47.