

Microwave dielectric properties of BaV₂O₆ ceramics with ultra-low sintering temperature

Cui Jin Pei^{a,b}, Guog Guang Yao^{a,b,*} and Zhao Yu Ren^b

^aInstitute of Photonics & Photon-Technology, Northwest University, Xi'an 710069, China

^bSchool of Science, Xi'an University of Posts and Telecommunications, Xi'an 710121, China

Microwave dielectric properties of BaV₂O₆ ceramics were investigated for the first time. The BaV₂O₆ ceramic was prepared by the solid-state ceramic route. The phase constitution, microstructure, and compatibility with aluminum electrode were investigated using XRD, Raman spectra and SEM. The results confirmed that that pure phase BaV₂O₆ could be obtained in the sintering temperature range of 525 °C–600 °C. With an increase in the sintering temperature, the dielectric constant (ϵ_r) and quality factor (Qxf) first increase and decrease thereafter, the temperature coefficient of resonant frequency (τ_f) changes slightly. The BaV₂O₆ ceramic sintered at 575 °C/4 hr exhibited good microwave dielectric properties of $\epsilon_r = 11.5$, $\tau_f = 38$ ppm/°C and Qxf = 21 800 GHz (at 10.1 GHz). Moreover, the material had a chemical compatibility with aluminum powders, which represented a promising candidate material for ULTCC applications.

Key words: Dielectric properties, BaV₂O₆, sintering.

Introduction

Intensive development of wireless communication systems has resulted in an increasing demand for microwave dielectric materials with low dielectric constant (ϵ_r), high quality factor (low dielectric loss, Qxf > 10 000 GHz) and a near-zero temperature coefficient of resonant frequency (τ_f) [1]. In addition, it is desired to integrate the passive components to a function module by low temperature co-fired ceramic (LTCC) technology, which requires the microwave dielectric ceramics to have lower sintering temperature than that of the highly conductive and inexpensive internal electrodes such as silver and aluminum. Most of the reported LTCC material systems use glass or oxide additives to reduce their sintering temperatures by negotiating their dielectric properties [2–4]. Therefore, the search for genuine low firing ceramics has always attracted much attention.

Recently extensive efforts were made in V₂O₅-based compounds to identify novel microwave ceramic systems suitable for LTCC substrate application since they possess low or even ultra-low sintering temperature and excellent microwave dielectric properties [5–9]. In BaO–V₂O₅ binary system, four phases are formed: Ba₃(VO₄)₂, Ba₂V₂O₇, Ba₃V₄O₁₃ and BaV₂O₆ [10]. Among these compounds, the microwave dielectric properties of Ba₃(VO₄)₂, Ba₂V₂O₇ and Ba₃V₄O₁₃ ceramics have been investigated [11–13].

They show good dielectric properties with ϵ_r of about 9.6–17, Qxf value in the range of 51 630–62 350 GHz, and δ_f ranging from –42 ~ 40 ppm/°C. Ba₂VO₆ has orthorhombic structure with space group (C222), and its photo-catalytic property has been reported recently [14, 15]. However, its microwave dielectric properties have not been investigated to date. Moreover, since the melting point of this ceramic is about 700 °C [10], the BaV₂O₆ compound is considered to be an appropriate candidate for new ultra-low temperature co-fired ceramic (ULTCC) material. In present study, the sintering behavior, microwave dielectric properties, and chemical compatibility with Al of BaV₂O₆ ceramic are introduced as new ULTCC materials.

Experimental

Samples of the BaV₂O₆ ceramic was synthesized by the conventional solid-state methods using high-purity oxide powders (> 99%): BaCO₃ and V₂O₅. The pre-dried raw materials were mixed according to the desired stoichiometry of the BaV₂O₆ ceramics. The powders were ground for 7 hrs in a nylon jar with agate balls and ethanol as media. All mixtures were dried and calcined at 460 °C for 3 hrs. The calcined powders were reground for 7 hrs, dried, mixed with 5 wt% PVA as a binder, and granulated. The granulated powders were uniaxially pressed into pellets with 10 mm in diameter and 4–5 mm in height under the pressure of 200 MPa. These pellets were sintered from 500 °C to 600 °C for 4 hrs in air with a heating rate of 5 °C/min.

The bulk densities of the sintered ceramics were measured by Archimedes' method. The crystal structures

*Corresponding author:
Tel : +86-29-88166089
Fax: +86-29-88166333
E-mail: yaoguoguang@xupt.edu.cn

were analyzed using X-ray powder diffraction (XRPD) with Cu K α radiation (Rigaku D/MAX2550, Japan). The Raman spectra were collected at room temperature using a Raman microscope (Horiba Jobin Yvon S.A.S., France) with Nd: YAG laser (532 nm) operated at 30 mw. The microstructure of pellets was investigated using a scanning electron microscope (SEM, Fei Quanta 200, Holland). The microwave dielectric properties of sintered samples were measured using a network analysis (ZVB20, Rohde & Schwarz, Germany) with the TE₀₁₈ shielded cavity method. The temperature coefficient of resonant frequency (τ_f) was calculated with the following equation:

$$\tau_f = \frac{f_{80} - f_{20}}{f_{20} \times (80 - 20)} \quad (1)$$

Results and Discussion

Fig. 1 shows the XRPD patterns of BaV₂O₆ ceramics sintered at different temperatures for 4 hrs. In the sintering temperature range of 525 °C–600 °C, pure phase BaV₂O₆ was obtained within the detectable level of XRPD. Locations of diffraction peak in XRPD patterns of the sintered ceramics did not vary significantly with increasing sintering temperature. The lattice parameters of BaV₂O₆ sintered at 575 °C were calculated as $a = 8.5116$ Å, $b = 12.4698$ Å, and $c = 7.8639$ Å with the Jade 6.0 software.

In order to further study the crystal structure of BaV₂O₆ in detail. The room-temperature Raman spectra of BaV₂O₆ ceramics sintered at 575 °C had been recorded and the result is presented in Fig. 2. The vibration modes in the Raman spectra of the BaV₂O₆ ceramics are classified into two groups, internal and external. Yao et al. reported that the crystal structure of BaV₂O₆ consists of chains of vertex-shared VO₄ tetrahedra along the <120> direction and Ba atoms located interstitially between the chains [14]. Thus, the internal vibrations are related to the [VO₄]³⁻ molecular group with a stationary mass center, the external vibrations or lattice phonons are associated with the motion of the Ba²⁺ cations and rigid molecular units [16]. The strong Raman bands observed in the 967–874 cm⁻¹ can be assigned to the symmetric stretching vibration of (VO₄)³⁻ ions. The symmetric bending vibrations of (VO₄)³⁻ anions are observed in the region between 470 and 300 cm⁻¹, whereas that of asymmetric bending vibration is observed at 527 cm⁻¹. The Raman bands below 300 cm⁻¹ are attributed to the external modes, which consist of the translational modes of the Ba²⁺, and (VO₄)³⁻ ions as well as librations of (VO₄)³⁻ ions [17].

The SEM photographs of as-fired surface in BaV₂O₆ specimens sintered at various temperatures are illustrated in Fig. 3. The BaV₂O₆ sample sintered at 525 °C showed a distribution of grain sizes smaller than 2 μ m and an appreciable amount of intergranular

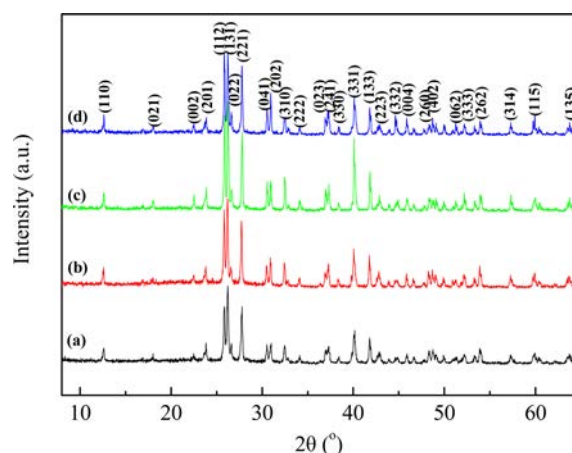


Fig. 1. XRPD patterns of BaV₂O₆ ceramics sintered at different temperatures for 4 hrs; (a) 525 °C, (b) 550 °C, (c) 575 °C, and (d) 600 °C.

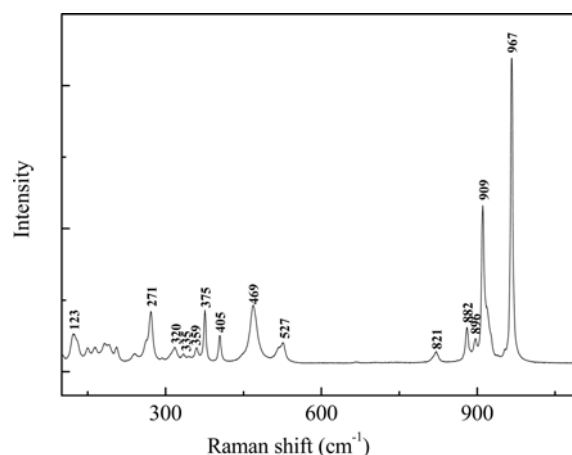


Fig. 2. The room-temperature Raman spectra of BaV₂O₆ ceramics sintered at 575 °C.

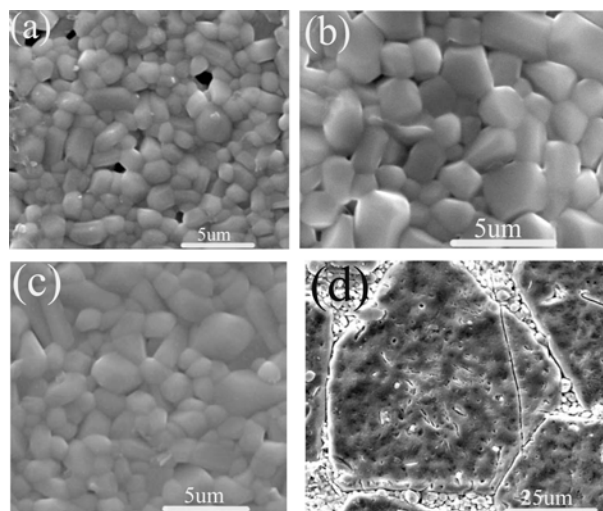


Fig. 3. SEM images of as-fired surface in BaV₂O₆ ceramics sintered at (a) 525 °C, (b) 550 °C, (c) 575 °C, and (d) 600 °C for 4 hrs.

porosity, as seen in Fig. 3(a). The pores decreased with increasing sintering temperature. When the sample was

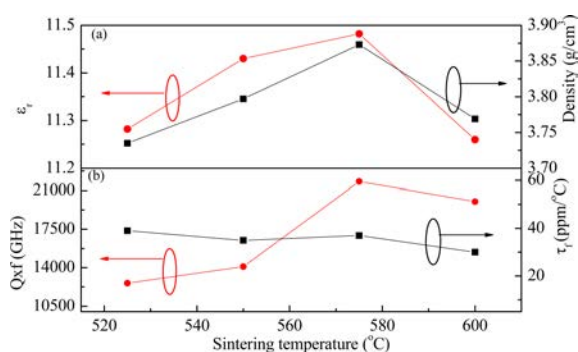


Fig. 4. Variation in (a) ϵ_r and density, and (b) Q_{xf} and τ_f of BaV_2O_6 ceramics versus firing temperatures.

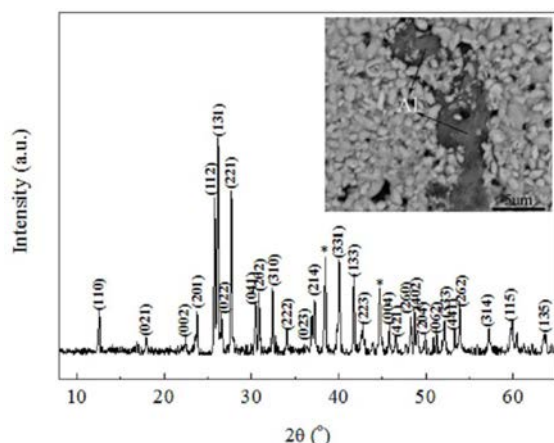


Fig. 5. XRD patterns and SEM image of $\text{BaV}_2\text{O}_6 + 20 \text{ wt}\% \text{ Al}$ sintered at 575°C for 4 hrs.

sintered at 575°C , the dense microstructures with almost no porosity could be observed, as seen in Fig. 3(c). However, abnormal grain growth was observed for BaV_2O_6 ceramic after sintering at 600°C , which would deteriorate the density and microwave dielectric properties of specimens.

Fig. 4 illustrates the density and microwave dielectric properties of BaV_2O_6 ceramics as a function of sintering temperature. The density of the samples increased with ascending the sintering temperature. After reaching a maximum value at 575°C , the density decreased due to the abnormal grain growth. As seen from Fig. 4(a), as the sintering temperature increased, ϵ_r increased with a trend similar to their density, because both the density and ϵ_r of ceramic are associated with the elimination of the pores [18]. In general, the dielectric loss is affected by intrinsic factors such as structural characteristics and extrinsic factors like secondary phase, grain size, densification or porosity [19, 20]. As shown in Fig. 4, the variation of Q_{xf} is commensurate with density, suggesting the loss mechanism might be dominated by extrinsic contributions. The Q_{xf} value increased with increasing temperature up to 575°C , which is attributed to the increase in density and grain size. After 575°C , the Q_{xf} value decreased due to the abnormal grain growth, as shown in Fig. 3(d). The τ_f values did not change

remarkably with increasing sintering temperature and remained stable at about $38 \text{ ppm}/^\circ\text{C}$. It is to be noted that few dielectric ceramics owing low ϵ_r , positive τ_f and intrinsic low sintering temperature has been reported to date [21]. In general, low ϵ_r materials possess large negative τ_f [22,23], however, the BaV_2O_6 ceramic exhibits an exceptionally positive τ_f and low ϵ_r . The positive τ_f and low ϵ_r of BaV_2O_6 is potentially attractive for its use as an end-member in composite ceramics with near-zero τ_f for substrate application.

To evaluate the chemical compatibility of BaV_2O_6 ceramic with aluminum electrodes, mixtures of the BaV_2O_6 powders and 20 wt% Al powders were cofired at $575^\circ\text{C}/4 \text{ hrs}$. The XRPD pattern and SEM image of the cofired sample are presented in Fig. 5. As the XRD pattern does not show the formation of any secondary phase except for BaV_2O_6 and Al, and the SEM image reveals no reaction boundary layer was detected between BaV_2O_6 and Al. These results show that the material under study has a chemical compatibility with Al.

Conclusions

A new V-based microwave dielectric ceramic BaV_2O_6 has been prepared and characterized. Pure orthorhombic phase ceramics could be obtained by the conventional solid-state route after sintering at $525\text{--}600^\circ\text{C}$. The BaV_2O_6 ceramics sintered at 575°C for 4 hrs showed good microwave dielectric properties of $\epsilon_r = 11.5$, $\tau_f = 38 \text{ ppm}/^\circ\text{C}$ and $Q_{xf} = 21\,800 \text{ GHz}$ (at 10.1 GHz), and it did not react with Al electrode. The unusually lower ϵ_r , large positive τ_f and reasonable values of Q_{xf} in BaV_2O_6 coupled with its low sintering temperature suggested that it may be useful in the fabrication of temperature-stable composites with lower ϵ_r , negative τ_f and high Q_{xf} end members.

Acknowledgments

This work is supported by the National Natural Science Foundation of China (Grants Nos 51402235, 51272150, and 51132003) and the Natural Science Foundation of Shaanxi Province, China (Grant No. 2015JQ5143). China's Post-doctoral Science Fund.

References

1. D. Zhou, C.A. Randall, H. Wang, Xi Yao, J. Am. Ceram. Soc. 94 (2011) 348-350.
2. A. Kan, T. Moriyama, H. Ogawa, Jpn. J. Appl. Phys. 51 (2012) 09LF1-4.
3. J.J. Bian, J.Y. Wu, L. Wang, J. Eur. Ceram. Soc. 32 (2012) 1251-1259.
4. A. Feteiraw, D.C. Sinclair, J. Am. Ceram. Soc. 91 (2008) 1338-1341.
5. E.K. Suresh, A.N. Unnimaya, A. Surjith, R. Ratheesh, Ceram. Int. 39 (2013) 3635-3639.

6. L. Fang, C.X. Su, H.F. Zhou, H. Zhang, *J. Am. Ceram. Soc.* 96 (2013) 688-690.
7. G.G. Yao, P. Liu, H.W. Zhang, *J. Am. Ceram. Soc.* 96 (2013) 1691-1693.
8. H.F. Zhou, F. He, L. Fang, *Ceram. Int.* 40 (2014) 6335-6338.
9. G.G. Yao, P. Liu, J.J. Zhou, H.W. Zhang, *J. Eur. Ceram. Soc.* 34 (2014) 2983-2987.
10. A.A. Fotiev, V.V. Makarov, V.V. Volkov, L. L. Surat, *Russian J. Inorg. Chem.* 14 (1969) 144-146.
11. R. Umemura, H. Ogawa, A. Yokoi, H. Ohsato, A. Kan, *J. Alloy. Compd.* 424 (2006) 388-393.
12. M.R. Joung, J.S. Kim, S. Nahm, *J. Am. Ceram. Soc.* 93 (2010) 934-936.
13. S.E. Kalathil, U.A. Neelakantan, R. Ratheesh, *J. Am. Ceram. Soc.* 97 (2014) 1530-1533.
14. T. Yao, Y. Oka, N. Yamamoto, *Inorg. Chim. Acta.* 238 (1995) 165-168.
15. M. Wang, Q. Liu, H.Y. Luan, *Adv. Mater. Res.* 460 (2012) 131-134.
16. C.S. Lim, *J. Ceram. Process Res.* 13 (2012) 432-436.
17. A. Grzechnik, P.F. McMillan, *J. Solid. State. Chem.* 132 (1997) 156-162.
18. X.G. Wu, H. Wang, Y.H. Chen, D. Zhou, *J. Am. Ceram. Soc.* 95 (2012) 1793-1795.
19. X.Y. Chen, S.X.M. Li, W.J. Zhang, *J. Eur. Ceram. Soc.* 33 (2013) 3001-3006.
20. K.M. Manu, T. Joseph, M.T. Sebastian, *Mater. Chem. Phys.* 133 (2012) 21-23.
21. A.N. Unnimaya, J. Dhanya, R. Ratheesh, *J. Mater. Sci: Mater. Electron.* 25 (2014) 1127-1131.
22. X.Y. Chen, S.X. Bai, W.J. Zhang, *J. Alloy. Compd.* 541 (2012) 132-136.
23. I.M. Reaney, D. Iddles, *J. Am. Ceram. Soc.* 89 (2006) 2063-2072.