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Concentration quenching of Tb³⁺ luminescence in Tb_XA_{3-X}Al₅O₁₂ (A=Yb, Y) single crystals

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Tb-doped Yb₃Al₅O₁₂ and Y₃Al₅O₁₂ single crystals were grown by the micropulling-down method in the form of rods of 2 mm diameter and a few cm in length. Emission spectra and decay times of Tb³⁺ luminescent centers were measured for Tb concentrations in a Y(Yb) sublattice of (1-70) %. A noticeable decrease of the emission intensity and decay time was obtained for Tb concentrations above 15% and 30% for Y₃Al₅O₁₂ and Yb₃Al₅O₁₂ hosts, respectively. Both effects are ascribed to the concentration quenching of Tb³⁺ luminescence due to energy migration over the Tb³⁺ subsystem and subsequent energy transfer to unspecified nonradiative traps. Energy migration process and structural aspects are discussed.

Key words: micropulling-down crystal growth, luminescence, rare earth ions, concentration quenching.

Introduction

Rare earth (RE) ion-doped single crystals of Y₃Al₅O₁₂ (YAG) garnet have been studied in many papers due to their extensive use for solid state lasers (Nd-doped [1]), scintillator (Ce-doped [2]) and other applications. For such applications usually dopant concentration up to a few percent is used to avoid unwanted effects arising in the case of high RE ion concentration like crossrelaxation, concentration quenching, etc. For RE ions similar in size to the Y cation, it is easy to prepare solid solutions of a mixed-cation garnet and/or to grow even the rare earth aluminium garnets, which are of the same structure, e.g. Tb₃Al₅O₁₂ (TbAG) [3] or Yb₃Al₅O₁₂ (YbAG) [4]. Another mixed Tb₃Sc₂Al₃O₁₂ garnet single crystal growth was reported recently mentioning its possible application as a Faraday rotator [5]. Tb^{3+} luminescence based on radiative transitions within the 4f-shell is well-known in many host materials [6] showing typically two emission multiplets related to the radiative transitions from the excited state ${}^{5}D_{4}$ level (dominating lines at 486 nm, 542 nm, 580 nm and 620 nm) and ⁵D₃ level (lines within 370-480 nm spectral region) to the ${}^{7}F_{x}$ ground state manifold. The ${}^{5}D_{3} \rightarrow {}^{7}F_{x}$ radiative transitions are strongly suppressed in the emission spectra for Tb³⁺ concentrations above a few percent due to cross-relaxation processes between two closely spaced Tb³⁺ ions.

It is the aim of this paper to report the concentration

quenching effects in Tb^{3+} luminescence, which occur in parallel to the above mentioned cross-relaxation process for higher Tb concentrations in YbAG and YAG hosts. While a cross-relaxation process does not result in the overall emission quenching (it is "just" an energy conversion from the ⁵D₃ to ⁵D₄ excited state), concentration quenching brings substantial emission intensity loss due to energy migration over the Tb-subsystem and its dumping in nonradiative traps.

Experimental

Single crystals of $Tb_xA_{3-x}Al_5O_{12}$ (A=Y or Yb) with varying concentrations of Tb^{3+} cations were grown in the form of rods (2 mm in diameter) and a few cm in length by the micropulling-down method. Rods were cut into 8 mm pieces and flattened to 0.8 mm thickness with optically polished surfaces.

The spectra and decay of Tb^{3+} photoluminescence were measured at room temperature (RT) in a Spectrofluorometer 199S (Edinburgh Instruments) using a steadystate hydrogen or microsecond pulse xenon flashlamp (IBH Scotland), respectively, as an excitation source. Photoluminescence spectra were corrected for spectral distortions and the decays were approximated typically by a single exponential function to extract the decay times (DT).

Experimental Results and Discussion

In the emission spectra of Tb-containing YbAG or YAG matrices a typical four-line spectrum peaking at about 488 nm, 542 nm, 584 nm and 621 nm was

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Fig. 1. Excitation (a) and emission (b) spectra of Tb^{3+} centers in YbAG:Tb (5%) at room temperature. (a) em= 542 nm; (b) ex= 325 nm.

obtained, which is ascribed to the radiative transition from the ⁵D₄ excited state of Tb³⁺ to the ground ⁷F_{6.5.4.3} multiplet – Fig. 1. In the excitation spectrum of the 542 nm emission line in Fig. 1 the dominant peaks at about 488 nm, 372+379 nm and 354 nm can be ascribed to the upward transitions from the ${}^{7}F_{6}$ ground state to the ${}^{5}D_{4}$, ${}^{5}D_{3}$ and ${}^{5}D_{2}$ excited state levels, respectively. With an increasing concentration of Tb³⁺, a considerable decrease of emission intensity and shortening of the emission decay time was obtained, see Fig. 2 for Tb_{2.2}Yb_{0.8}AG. Such behavior is typical, when the luminescence concentration quenching occurs, i.e. when the luminescence ions become so close that the excitation energy starts to migrate among them and becomes progressively lost at unspecified nonradiative traps (lattice defects) [7]. The concentration dependence of decay times of the Tb³⁺ 542 nm emission line is shown in Fig. 3 for both YAG and YbAG matrices. It can be seen that the knee on the curves is obtained above 30% of Tb³⁺ in the case of the YbAG matrix, while a some-



Fig. 2. Photoluminescence decay of $Tb_{2.2}Yb_{0.8}Al_5O_{12}$ at room temperature. Exc=490 nm, em=542 nm. A single exponential approximation is given by the solid line. The decay time of 0.096 ms was calculated.

what lower concentration between 15-30% is enough to obtain noticeable DT shortening in the YAG matrix. Furthermore, a higher isolated Tb^{3+} DT (low concentration limit) of about 3 ms was obtained in YAG compared to about 1 ms in YbAG matrix.

Interestingly enough, the decay curves do not deviate from a single exponential course for any Tb concentration (example in Fig. 2), which is expected in the case of the multipolar interaction between the donor themselves (Tb³⁺ ion) and donor-acceptor (a nonradiative trap) pairs. Diffusion-limited energy transfer processes are theoretically described for multipolar interactions in generally accepted energy transfer models (e.g. Yokota-Tanimoto limited diffusion model [8], Ghosh-Hegarty-Huber model [9]), which show wellexpressed non-exponentialities in the initial part of the decay curves. Due to the fact that the emission transition of Tb³⁺ occurs between the well-screened 4f-like levels and it is strongly forbidden (decay times of 1-3 ms), the interaction between Tb^{3+} ions and between Tb³⁺ and nonradiative traps could be considered to be of short-range exchange character and a relatively high Tb³⁺ concentration is thus necessary to achieve a significant concentration quenching effect. In the case of exchange interaction the energy transfer probability decreases exponentially with the Tb-Tb (Tb-trap) distance and above 6-10 Å it becomes negligibly small [10]. In such a case the number of donor ions in the sphere of the influence of an acceptor will always be very small (see also below) and the hopping model of Burshtein seems to be more appropriate [11], in which the above mentioned non-exponential decay features may become less significant. It follows from the above text that whatever is the nature of the acceptor nonradiative traps, their interaction with Tb³⁺ donor subsystems must be of very short (probably exchange) character.



To provide more insight into structural features of

Fig. 3. Tb^{3+} concentration dependencies of the decay times related to the 542 nm emission line of the Tb³⁺ emission center in YbAG and YAG matrices measured under exc=490 nm and room temperature.



Fig. 4. Sketch of TbAG structural arrangement along the b axis with a central Tb^{3+} ion (oxygen anions omitted).

(Tb-Yb)AG and related possibilities for the energy migration among Tb³⁺ ions, the number of Tb ions within a sphere of 10 Å from a central Tb ion is calculated together with from-central-Tb ion distances by considering TbAG structure [4]. A sketch of the cation spatial arrangement is given in Fig. 4. We obtain the result that from 51 Tb ions present in such a sphere, there are four ions at a distance of 3.67 Å, eight ions at a distance of 5.61 Å and two ions at a distance of 6.00 Å from the central Tb ion. As the exchange interaction probability between Tb-Tb pairs decreases exponentially with their mutual distance, the energy transfer probability will decrease 5-6 times, if transfer from the central Tb^{3+} ion over the 5.61 Å distance to the next-nearest Tb³⁺ ions is considered, with respect to the closest (3.67 Å) Tb³⁺ cations. An abrupt decrease in decay time in Fig. 3 for Tb concentrations above 15% (YAG) and 30% (YbAG) must be thus connected with the increased occurrence of Tb3+ ions at the nearest neighbor (3.67 Å) positions to allow efficient energy migration over possibly many Tb³⁺ ions until a nonradiative trap has been reached. Deeper quantitative considerations are excluded, however, as the nature and concentration of the acceptor nonradiative traps are unknown.

Based on these simple considerations, it is however difficult to understand the reason for the earlier (lower Tb concentration) onset of the decay time shortening in the YAG matrix, where Y^{3+} cations are of even bigger size with respect to the Yb³⁺ ones. A bigger cation implies generally a bigger lattice constant for the structure, which could in principle make the diffusion over a Tb³⁺-doped subsystem a little less efficient. This aspect thus needs further consideration. Furthermore, a significant difference in the low concentration limit of the Tb³⁺ decay times in YAG and YbAG matrices, which is probably related to different mixing Tb³⁺ 4f states with the opposite parity states of the host, would deserve a deeper consideration as well.

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